

# Preparation and structure of copper nanoparticle/poly(acrylic acid) composite films

Yasuo Gotoh,<sup>\*a</sup> (the late) Ryo Igarashi,<sup>a</sup> Yutaka Ohkoshi,<sup>a</sup> Masanobu Nagura,<sup>a</sup> Kensuke Akamatsu<sup>b</sup> and Shigehito Deki<sup>b</sup>

<sup>a</sup>Faculty of Textile Science and Technology, Shinshu University, 3-15-1 Ueda, Nagano 386-8567, Japan. E-mail: ygotohy@giptc.shinshu-u.ac.jp

<sup>b</sup>Department of Chemical Science & Engineering, Faculty of Engineering, Kobe University, 1-1 Rokkodai, Nada, Kobe, 657-8501, Japan

Received 16th May 2000, Accepted 8th August 2000

First published as an Advance Article on the web 21st September 2000

Composite films consisting of metallic Cu nanoparticles dispersed in poly(acrylic acid) (PAA) have been prepared by reduction of Cu<sup>2+</sup> from the copper salt of PAA above 220 °C under a H<sub>2</sub> atmosphere. Optical absorption properties and structures have also been investigated by UV/VIS, WAXD, TEM and IR. Spherical Cu particles were found to be homogeneously dispersed in the PAA and the diameters of the particles were in the range 10–16 nm. The composite films exhibited an optical absorption peak centered at *ca.* 570 nm, which was attributed to surface plasmon resonance of Cu nanoparticles. The composite film made by heat treatment at 220 °C was less stable because Cu particles in the film were oxidized to Cu<sup>2+</sup> ions within several weeks, while the composite films prepared by heating above 230 °C were stable and the Cu particles in their films were not oxidized. The stability of the Cu nanoparticles in PAA is suggested to be related to the formation of ketone groups by condensation reactions between carboxylic acids of PAA above 230 °C.

## Introduction

Recently, numerous studies have been carried out on composite materials consisting of glasses or polymers doped with metal and semiconductor nanoparticles because of the attraction of their optical properties, *e.g.*, nonlinear optical and laser amplification properties.<sup>1–5</sup> For metal particles in composites of organic polymers, most of the investigations have dealt with noble metals such as Au and Ag, because these nanoparticles are known to exhibit characteristic optical absorption caused by surface plasmon resonance (SPR) originated from collective oscillation of free electrons.<sup>6</sup> Various methods to prepare metal-doped polymer films have been developed and the characteristics of these films have been studied in detail by many researchers.<sup>1–4</sup> For materials to be utilized in future applications, it is important to ensure chemical and thermal stability of the composites and to control the size and concentration of the dispersed metal particles to give the desired properties for the obtained materials. One of the present authors has previously prepared polymer thin films containing noble metal nanoparticles by a thermal relaxation technique, and the relationship between the properties and the film microstructure has been investigated.<sup>7–11</sup>

In contrast to noble metal nanoparticles, few studies have so far been carried out on polymer composites doped with nanoparticles of non-noble metals such as Cu. Cu nanoparticles exhibit SPR absorption as well as Au and Ag, and there are many reports on the preparation of Cu particles embedded in inorganic glass matrices and their optical properties.<sup>12–16</sup> In general, nanoparticles in inorganic glasses have been prepared by heating at relatively high temperatures. Since high temperature annealing generally causes the thermal degradation of organic polymers, it is difficult to prepare Cu nanoparticles in organic polymer matrices by heating at high temperatures. In this study we report on a new procedure for the preparation of Cu metal nanoparticle/polymer composite films by heat treatment. Namely, composite films consisting of

Cu nanoparticles and poly(acrylic acid) (PAA) have been prepared by the reduction of Cu<sup>2+</sup> from the copper salt of PAA over 200 °C under H<sub>2</sub>, and the optical absorption properties and structures of the prepared composites have also been investigated.

## Experimental

### Materials

A 25 wt% PAA aqueous solution was purchased from Wako Pure Chemical Industries and was used without any purification. The viscosity-average molecular weight of the PAA was *ca.* 150 000. Anhydrous copper(II) acetate was obtained from Yoneyama Chemical Industries and was used without any purification.

### Preparation of Cu nanoparticle/PAA composites

The PAA aqueous solution was smeared on a glass plate which was then immersed in a 5 wt% copper(II) acetate aqueous solution at room temperature for 5 min. A gel film was immediately formed, which was blue-green and insoluble in water. This indicates that the PAA was cross-linked by coordination of the carboxylic acids of PAA to Cu<sup>2+</sup>. When 50 mg of the PAA aqueous solution was smeared on a glass plate measuring 26 × 55 mm in area, a gel film of *ca.* 10 μm thickness was obtained. PAA (1 g) reacted with 0.19 g Cu<sup>2+</sup> under the present preparation conditions. After drying, the PAA–Cu<sup>2+</sup> films were heated at various temperatures above 200 °C for 30 min under a H<sub>2</sub> atmosphere so as to obtain Cu nanoparticle/PAA composites. In this case H<sub>2</sub> was used as a reducing agent of Cu<sup>2+</sup>.

### Measurements

Infrared spectroscopy (IR) measurements were performed with a Perkin-Elmer FT-IR spectrometer 1640. Measurement

conditions were 64 scans and a resolution of  $2\text{ cm}^{-1}$ . Electron spin resonance (ESR) measurements were carried out with a JEOL X-band ESR spectrometer TE200 at  $23\text{ }^\circ\text{C}$ . Ultraviolet-visible (UV/VIS) absorption spectra were measured using a Hitachi UV/VIS/NIR spectrophotometer U-3500. Wide angle X-ray diffraction (WAXD) experiments were performed on a Rigaku Geigerflex 2028 diffractometer with  $\text{CuK}\alpha$  radiation. The voltage and current of the X-ray source were 40 kV and 150 mA, respectively. Crystallite sizes were calculated from the  $\text{Cu}(111)$  diffraction line using Scherrer's equation,  $L = \lambda/(\beta \cdot \cos\theta)$ , where  $L$  is the mean dimension of the crystallites,  $\beta$  is the integral width of the diffraction peak,  $\theta$  is the diffraction angle, and  $\lambda$  is the wavelength of the  $\text{CuK}\alpha_1$  radiation ( $0.1540\text{ nm}$ ). The observed integral width of each diffraction peak was corrected for both the  $\text{CuK}\alpha$  doublet broadening and the instrumental broadening. Transmission electron microscopy (TEM) observation was performed with a JEOL JEM-2000 transmission electron microscope operated at 200 kV. Thin sections (*ca.* 100 nm in thickness) for TEM observation were obtained by the conventional microtome technique. Thermal mechanical analysis (TMA) was carried out with a SEIKO SS-10 at a heating rate of  $1\text{ }^\circ\text{C min}^{-1}$  under a  $\text{H}_2$  flow.

## Results and discussion

### Complexation between PAA and $\text{Cu}^{2+}$

In order to confirm the formation of the PAA copper salt and investigate the coordination state of  $\text{Cu}^{2+}$  in the specimen, IR and ESR measurements were carried out. Fig. 1 shows the IR spectra of the PAA and the  $\text{PAA-Cu}^{2+}$  films. The carbonyl asymmetric stretching band [ $\nu(\text{C}=\text{O})$ ] of unneutralized  $-\text{CO}_2\text{H}$  is found at  $1700\text{ cm}^{-1}$  in the spectrum of PAA. On the other hand, for the  $\text{PAA-Cu}^{2+}$  film the  $\nu(\text{C}=\text{O})$  of unneutralized  $-\text{CO}_2\text{H}$  is considerably weakened and three peaks appeared around  $1500\text{--}1600\text{ cm}^{-1}$ . Because shifts to lower wavenumbers are generally observed upon complexation,<sup>17</sup> these peaks can be assigned to the  $\nu(\text{C}=\text{O})$  band complexed with  $\text{Cu}^{2+}$ , and the peaks suggest that there are some different coordination states coexisting within the  $\text{Cu}^{2+}$  complexes.

Fig. 2 shows the ESR spectrum of  $\text{PAA-Cu}^{2+}$ . Two bands are observed around 310 mT and 455 mT, indicating that two coordination states of  $\text{Cu}^{2+}$  coexist. According to the  $g$ -values of the two bands, the former is an anisotropic band with

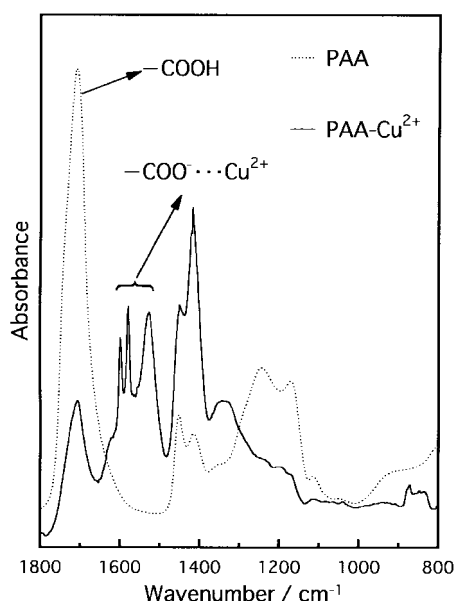


Fig. 1 FT-IR spectra of PAA and  $\text{PAA-Cu}^{2+}$  prepared by the reaction of PAA with copper(II) ion in the treatment solution.

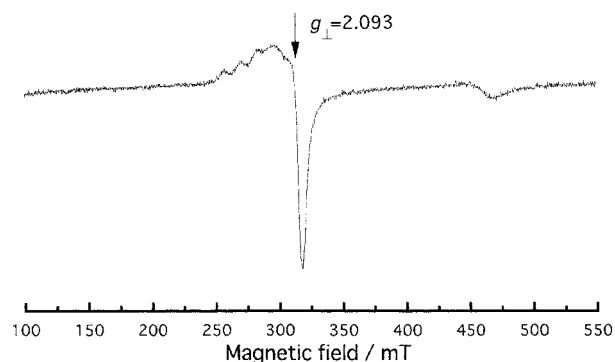


Fig. 2 ESR spectrum of  $\text{PAA-Cu}^{2+}$ , measured at  $23\text{ }^\circ\text{C}$ .

hyperfine structure and can be assigned to mononuclear  $\text{Cu}^{2+}$  complexes with four-oxygen ligation.<sup>18,19</sup> The latter, isotropic, band is assigned to a binuclear complex of the dimeric copper(II) acetate monohydrate type.<sup>19,20</sup> It is considered that most of the  $\text{Cu}^{2+}$  in  $\text{PAA-Cu}^{2+}$  is in the state of mononuclear  $\text{Cu}^{2+}$  type, because the relative intensity of the band at 310 mT is somewhat larger than that at 455 mT.

### Formation of Cu particles in PAA

Fig. 3 shows the UV/VIS spectra of the  $\text{PAA-Cu}^{2+}$  films heated under an  $\text{H}_2$  atmosphere at various temperatures. For the as-prepared  $\text{PAA-Cu}^{2+}$  film, the colour of the film was blue-green, and the UV/VIS spectrum shows an absorption peak centered at 690 nm, which is originated from the d-d transition of the 3d orbital in  $\text{Cu}^{2+}$ .<sup>21</sup> During heat treatment below  $210\text{ }^\circ\text{C}$  the colour of the  $\text{PAA-Cu}^{2+}$  films did not change, remaining blue-green similar to the as-prepared film. This indicates that  $\text{Cu}^{2+}$  ions in the films are not reduced by the heat treatment below  $210\text{ }^\circ\text{C}$ . The colour of the films significantly changed upon heating above  $215\text{ }^\circ\text{C}$ . The film heated at  $215\text{ }^\circ\text{C}$  became colourless and the spectrum showed no absorption in the visible region, indicating the absence of the d-d transition of the  $\text{Cu}^{2+}$  3d orbital. This is attributed to the reduction of  $\text{Cu}^{2+}$  to  $\text{Cu}^+$  by  $\text{H}_2$ . Consequently, the  $\text{Cu}^+$  3d orbital is filled with 10 electrons and the d-d transition does not occur. The absorption edge around 360 nm is probably caused by charge transfer between  $-\text{CO}_2^-$  and  $\text{Cu}^+$ . After heat treatment above  $220\text{ }^\circ\text{C}$ , the films changed to deep red and the spectrum shows an absorption peak at 565 nm. It is well known that metallic nanoparticles exhibit characteristic optical properties due to SPR absorption, which results in an absorption peak in the UV/VIS region.<sup>2,4,6,21-23</sup> In the present study the absorption peak at 565 nm is approximately

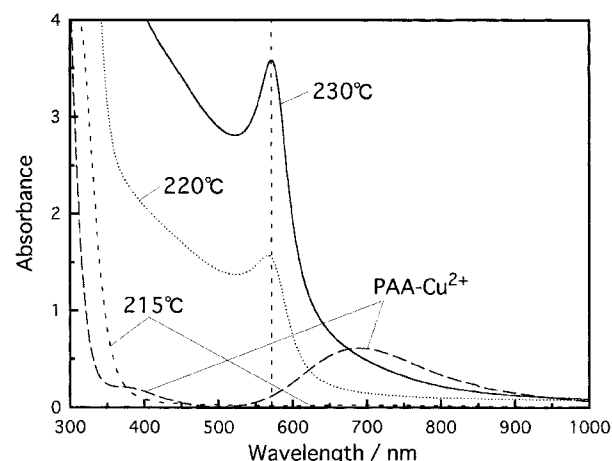


Fig. 3 UV/VIS spectra of the  $\text{PAA-Cu}^{2+}$  films heated at various temperatures under a  $\text{H}_2$  atmosphere for 30 min.

coincident with the SPR wavelength of Cu nanoparticles reported by different authors.<sup>11-15,24,25</sup> Therefore, it is considered that Cu nanoparticles were formed in the films by heat treatment above 220 °C. Upon heat treatment at 230 °C, an absorption peak is observed at 570 nm, the intensity is enhanced, and the position shifts slightly to longer wavelength in comparison with the PAA-Cu<sup>2+</sup> film heated at 220 °C. This suggests that the number of Cu particles formed in the matrix increases as the heat treatment temperature increases.<sup>7</sup>

From the results of UV/VIS the following reduction mechanism of Cu<sup>2+</sup> is proposed.

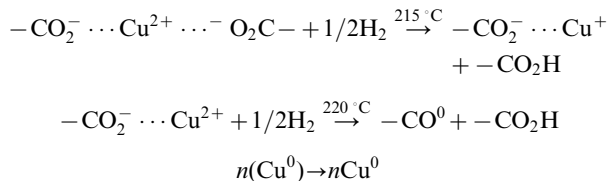


Fig. 4 shows the WAXD patterns of the films heat-treated at various temperatures under an H<sub>2</sub> atmosphere. PAA homopolymer is amorphous and exhibits a broad scattering pattern with a peak maximum at 2θ = ca. 17°. In Fig. 4, after complexation between Cu<sup>2+</sup> and PAA (as-prepared sample), the peak maximum shifts to ca. 20°. The WAXD pattern of the PAA-Cu<sup>2+</sup> heated below 210 °C is almost identical in shape to that of the as-prepared film and shows no diffraction peaks attributable to metallic Cu. This indicates that the Cu<sup>2+</sup> is not reduced by the heat treatment below 210 °C, and PAA-Cu<sup>2+</sup> remains unchanged. Upon heat treatment at 215 °C, the amorphous pattern is found to change although no diffraction peaks of metallic Cu are observed, i.e., the peak maximum of the amorphous scattering shifted from 20° to 23°. This change indicates a conformational change of the PAA molecular chains. This heat treatment temperature of 215 °C corresponds to that at which the reduction of Cu<sup>2+</sup> to Cu<sup>+</sup> occurs, as shown in Fig. 3. Therefore, the reduction to Cu<sup>+</sup> causes the changes in the conformation of the PAA molecules as well as the coordination state of copper ions. Upon heat treatment above 220 °C, a diffraction peak is observed at 43.3°, which is assigned to the (111) reflection of metallic Cu. Heat treatment at 230 °C results in an increase of the intensity of the diffraction peak for Cu metal and the line width is sharpened as compared to the film heated at 220 °C. In addition, the peak maximum of the scattering for PAA shifts to 17°, which is consistent with that of PAA homopolymer. These results indicate that most of

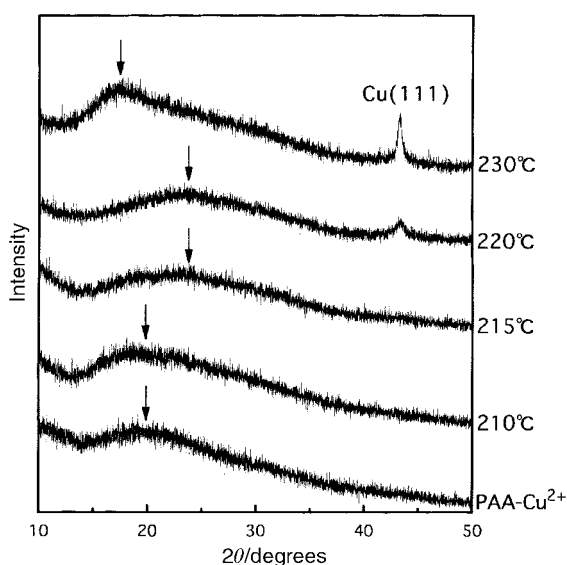


Fig. 4 WAXD patterns of the PAA-Cu<sup>2+</sup> films heated at various temperatures under a H<sub>2</sub> atmosphere for 30 min.

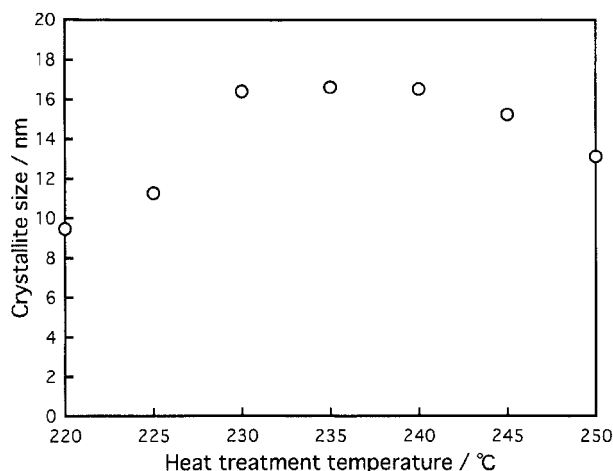


Fig. 5 Dependence of the crystallite sizes of Cu particles in PAA films on the heat treatment temperature.

Cu<sup>2+</sup> in the film is reduced to metallic Cu at 230 °C, which is in good agreement with the results of the UV/VIS measurement (Fig. 3). The crystallite size of the Cu particles in each composite film was calculated from the width of the Cu(111) diffraction line at 43.3°. Fig. 5 shows the dependence of crystallite size of the Cu particles on the heat treatment temperature. Each crystallite size is in the range of ca. 10–16 nm and small enough to exhibit SPR absorption of Cu particles.<sup>24,25</sup> The crystallite size increases with increasing heat treatment temperature up to 230 °C and is about 16 nm in the 230–240 °C range, and then slightly decreases again above 240 °C.

We observed directly the dispersion state of Cu particles in the film by TEM observation. Fig. 6 shows the TEM image of the cross-section for the film heat-treated at 230 °C. Spherical Cu particles are homogeneously dispersed in PAA; the particle diameter is 15–20 nm, which is nearly coincident with the crystallite size estimated by WAXD (*L* = 16.4 nm). We also attempted to prepare a thin section of the film heated at 220 °C for TEM observation, but this was unsuccessful. In preparing the ultrathin cross-sections for TEM observation with a microtome, distilled water is necessary to float them. However, the specimen was softened by swelling with distilled water and could not be cut into an ultrathin film. Furthermore, the colour of the film heated at 220 °C changed from red to blue-green within several weeks. This implies that Cu metal particles were oxidized and changed again to Cu<sup>2+</sup> in the film, i.e., the

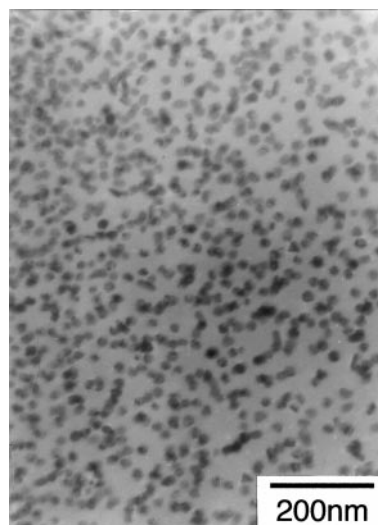


Fig. 6 Cross-sectional TEM image of Cu-PAA film heated at 230 °C under a H<sub>2</sub> atmosphere for 30 min.

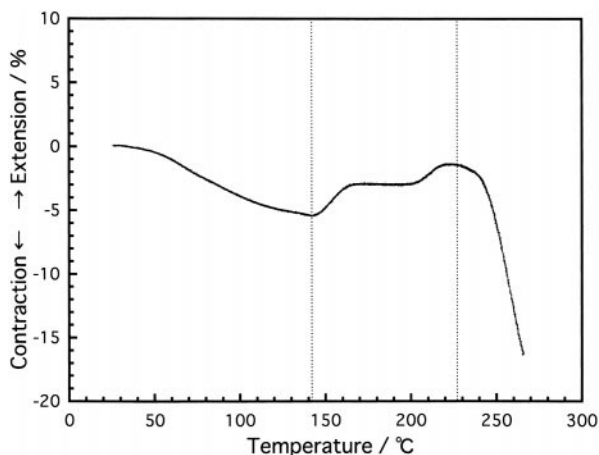


Fig. 7 TMA thermogram of PAA-Cu<sup>2+</sup> films under a H<sub>2</sub> flow.

composite film prepared by heat treatment at 220 °C is less stable. On the other hand, the films heat-treated above 230 °C hardly swelled in distilled water and the colour of the composites remained unchanged. Judging from these results, the intermolecular crosslinking between PAA chains is probably formed by heat treatment at 230 °C and the Cu particles can be stabilized in the matrix.

In order to investigate the formation of crosslinking in the specimens, TMA measurements were carried out. Fig. 7 shows the TMA curve of the as-prepared PAA-Cu<sup>2+</sup> film. Below 140 °C, the PAA-Cu<sup>2+</sup> film is observed to be gradually contracted with increasing temperature due to the elimination of water contained in the film, and then the film is extended stepwise up to 225 °C. This extension may be caused by softening of the matrix due to the glass transition temperature of PAA (*ca.* 103 °C<sup>27</sup>). The specimen considerably contracts again above 230 °C. It seems likely that this second contraction is attributed to a stiffening of the matrix, originated by the formation of intermolecular crosslinking between the PAA chains. Therefore, the film heated at 230 °C does not swell when the film is immersed in distilled water.

IR measurements were carried out to elucidate the binding formula of the intermolecular crosslinking between PAA chains in the film. Fig. 8 shows the IR spectra of the PAA and the PAA-Cu<sup>2+</sup> films heated at 220 °C and 230 °C, respectively. For the film heated at 220 °C, an absorption band which presumably may be assigned to the ν(C=O) band of -CO<sub>2</sub><sup>-</sup>...Cu<sup>+</sup> is observed at 1540 cm<sup>-1</sup>. Thus, it is reasonable

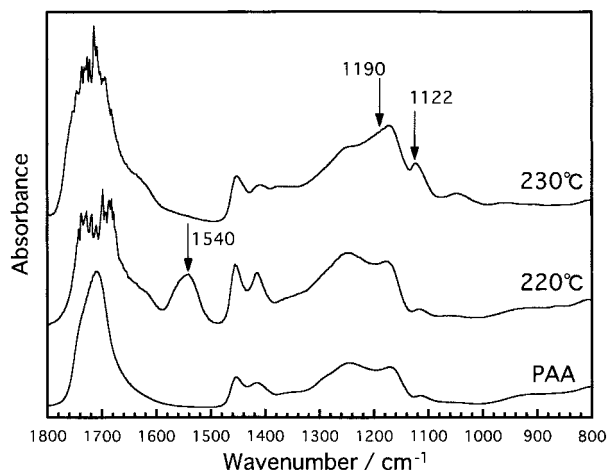
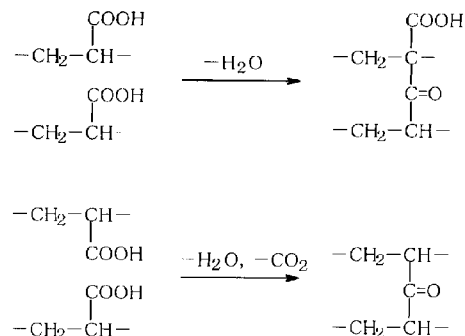


Fig. 8 IR spectra of PAA and PAA-Cu<sup>2+</sup> films heated at various temperatures.

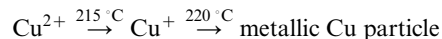
that all the Cu ions are not reduced by the heat treatment at 220 °C, as concluded from the results of UV/VIS and WAXD measurements. During heating of PAA above 150 °C crosslinking reaction between the molecular chains occurs.<sup>28</sup> It is well known that acid anhydride is formed by dehydration between -CO<sub>2</sub>H groups as a type of crosslinking.<sup>26,28</sup> For the film heated at 230 °C, the absorption band assigned to acid anhydride is not observed in the spectrum because of the absence of any absorption peak at *ca.* 1800 cm<sup>-1</sup>. Instead, two new absorption peaks are observed around 1100–1200 cm<sup>-1</sup>, which can be assigned to C-C stretching bands of C-C=O of ketone groups. The ketone groups are probably generated by condensation reaction between -CO<sub>2</sub>H groups of the PAA and bring about the crosslinking of PAA molecules. The possible condensation reaction routes for the formation of the ketone groups are expressed in the following equations.



As a result, the number of ketone groups instead of carbonyl groups increases and crosslinking between the PAA chains proceeds. The ketone does not react with Cu metal, therefore Cu nanoparticles are stable in the matrix. However, the PAA-Cu<sup>2+</sup> film heated at 220 °C exhibited no absorption bands of ketone groups, indicating that the carboxyl groups of PAA remain in the film. Consequently, the carboxyl groups react with Cu nanoparticles which lead to the formation of a -CO<sub>2</sub><sup>-</sup>-Cu<sup>2+</sup> complex, and, therefore, the colour of the film heated at 220 °C changes from red to blue-green as mentioned above. These results demonstrate that the formation of the crosslinking between PAA chains plays an important role in the stabilization of Cu nanoparticles in the matrix.

## Conclusion

Cu nanoparticle/PAA composite films have been prepared by the reduction of Cu<sup>2+</sup> from the copper salt of PAA at high temperature under H<sub>2</sub>. Upon thermal treatment, Cu<sup>2+</sup> in PAA-Cu<sup>2+</sup> was reduced by H<sub>2</sub> as follows.



The colour of the PAA-Cu<sup>2+</sup> films heated above 220 °C changed from blue-green to deep red and an optical absorption at *ca.* 570 nm was observed, caused by SPR absorption of Cu nanoparticles. Cu particles dispersed homogeneously in the films and the diameters were in the range 10–16 nm. The stability of the Cu particles in the composite films obtained above 230 °C was higher than that obtained at 220 °C. The difference of the stability of the Cu nanoparticles can be related to the formation of ketone groups by condensation reaction between the carboxylic acid groups of PAA above 230 °C. Namely, -CO<sub>2</sub>H groups in PAA-Cu<sup>2+</sup> remained when heated at 220 °C and reacted with Cu particles. Consequently, -CO<sub>2</sub><sup>-</sup>-Cu<sup>2+</sup> formed again and the colour of the composite reverted to blue-green. On heating above 230 °C ketone groups were generated in the composites by condensa-

tion reaction between  $-\text{CO}_2\text{H}$  groups, which are responsible for the stabilization of the Cu nanoparticles.

## References

- 1 Y. N. Cheong Chan, R. R. Schrock and R. E. Cohen, *J. Am. Chem. Soc.*, 1992, **114**, 7295.
- 2 Y. Nakao, *J. Chem. Soc., Chem. Commun.*, 1993, 826.
- 3 S. Ogawa, Y. Hayashi, N. Kobayashi, T. Tokizaki and A. Nakamura, *Jpn. J. Appl. Phys.*, 1994, **33**, L331.
- 4 L. L. Beecroft and C. K. Ober, *Chem. Mater.*, 1997, **9**, 1302.
- 5 S. Wang and S. Yang, *Langmuir*, 2000, **16**, 389.
- 6 U. Krebig and L. Ganzel, *Surf. Sci.*, 1985, **156**, 678.
- 7 K. Akamatsu and S. Deki, *J. Mater. Chem.*, 1997, **7**, 1773.
- 8 K. Akamatsu and S. Deki, *Nanostruct. Mater.*, 1997, **8**, 1121.
- 9 K. Akamatsu and S. Deki, *J. Mater. Chem.*, 1998, **8**, 637.
- 10 S. Deki, K. Akamatsu, T. Yano, M. Mizuhata and A. Kajinami, *J. Mater. Chem.*, 1998, **8**, 1865.
- 11 S. Deki, K. Akamatsu, Y. Hatakenaka, M. Mizuhata and A. Kajinami, *Nanostruct. Mater.*, 1999, **11**, 59.
- 12 V. V. Truong and G. D. Scott, *J. Opt. Soc. Am.*, 1977, **67**, 502.
- 13 E. Anno, M. Tanimoto and T. Yamaguchi, *Phys. Rev. B*, 1988, **38**, 3521.
- 14 R. Doremus, S. Kao and R. Garcia, *Appl. Opt.*, 1992, **31**, 5773.
- 15 H. Hosono, Y. Abe and N. Matsunami, *Appl. Phys. Lett.*, 1992, **60**, 2613.
- 16 K. Uchida, S. Kaneko, S. Omi, C. Hata, H. Tanji, Y. Asahara, A. J. Ikushima, T. Tokizaki and A. Nakamura, *J. Opt. Soc. Am. B*, 1994, **11**, 1236.
- 17 S. Yano, N. Nagao, M. Hattori, E. Hirasawa and K. Tadano, *Macromolecules*, 1992, **25**, 368.
- 18 H. Yokoi, S. Kawata and M. Iwaizumi, *J. Am. Chem. Soc.*, 1986, **108**, 3358.
- 19 H. Tomita, T. Goto, S. Shimada and K. Takahashi, *Polymer*, 1996, **37**, 1071.
- 20 J. Yamauchi, S. Yano and E. Hirasawa, *Makromol. Chem., Rapid Commun.*, 1989, **10**, 109.
- 21 C. J. Ballhausen, in *Introduction to Ligand Field Theory*, McGraw-Hill, New York, 1962.
- 22 R. H. Doremus, *J. Chem. Phys.*, 1964, **40**, 2389.
- 23 C. G. Granquist and O. Hunderi, *Phys. Rev. B*, 1977, **16**, 3513.
- 24 A. Yanase and H. Komiyama, *Surf. Sci.*, 1991, **248**, 11.
- 25 I. Lisiecki and M. P. Pileni, *J. Am. Chem. Soc.*, 1993, **115**, 3887.
- 26 M. Nagura and H. Ishikawa, *Kobunshi Ronbunshu*, 1983, **40**, 517.
- 27 A. Eisenberg, T. Yokoyama and E. Sambalido, *J. Polym. Sci.*, 1963, **C16**, 3943.
- 28 M. C. McGaugh and S. Kottle, *Polym. Lett.*, 1967, **5**, 817.